(19) World Intellectual Property Organization International Bureau







(43) International Publication Date 22 November 2001 (22.11.2001)

PCT

(10) International Publication Number WO 01/88226 A2

(51) International Patent Classification7:

C25D

(21) International Application Number: PCT/EP01/04923

(22) International Filing Date: 27 April 2001 (27.04.2001)

(25) Filing Language:

English

(26) Publication Language:

English

(30) Priority Data:

PCT/EP00/04608 18 May 2000 (18.05.2000) EP
EP 00203304.1 25 September 2000 (25.09.2000) EP
EP 1200628.4 21 February 2001 (21.02.2001) EP

(71) Applicant (for all designated States except US): CORUS ALUMINIUM WALZPRODUKTE GMBH [DE/DE]; Carl-Spaeter-Strasse 10, 56070 Koblenz (DE).

(72) Inventors; and

(75) Inventors/Applicants (for US only): MOOIJ, Joop, Nicolaas [NL/NL]; Walstro 56, NL-1902 JP Castricum (NL). WITTEBROOD, Adrianus, Jacobus [NL/NL]; Vederkruid 8, NL-1991 HB Velserbrock (NL). WI-JENBERG, Jacques, Hubert, Olga, Joseph [NL/NL]; Sumatrakade 63, NL-1019 PH Amsterdam (NL). (74) Agent: HERMAN DE GROOT, Johan, Willem; Corus Technology BV, P.O. Box 10000, NL-1970 CA Umuiden (NL).

(81) Designated States (national): AE, AG, AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, BZ, CH, CN, CR, CU, CZ, DE, DK, DM, DZ, EE, ES, FI, GB, GD, GE, GH, GM, HR, HU, ID, IL, IN, IS, JP, KE, KG, KP, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MA, MD, MG, MK, MN, MW, MX, MZ, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, TZ, UA, UG, US, UZ, VN, YU, ZA, ZW.

(84) Designated States (regional): ARIPO patent (GH, GM, KE, LS, MW, MZ, SD, SL, SZ, TZ, UG, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), European patent (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE, TR), OAPI patent (BF, BJ, CF, CG, CI, CM, GA, GN, GW, ML, MR, NE, SN, TD, TG).

Published:

 without international search report and to be republished upon receipt of that report

For two-letter codes and other abbreviations, refer to the "Guidance Notes on Codes and Abbreviations" appearing at the beginning of each regular issue of the PCT Gazette.

A2

(54) Title: METHOD OF MANUFACTURING AN ALUMINIUM PRODUCT

(57) Abstract: The invention relates to a method of manufacturing an Al or alloy workpiece, comprising the steps of (a) providing an AlAl alloy workpiece, (b) pre-treating of the outersurface of the Al or Al alloy workpiece, and (c) plating a metal layer comprising nickel onto said outersurface of the pre-treated Al or Al alloy workpiece, and wherein during step (c) said metal layer comprising nickel is deposited by electroplating both nickel and bismuth using an aqueous bath comprising a nickel-ion concentration in a range of 10 to 100 g/l and a bismuthion concentration in the range of 0.01 to 10 g/l. The invention further relates to an aqueous plating bath for use in the method of this invention.

10

15

20

25

30

METHOD OF MANUFACTURING AN ALUMINIUM PRODUCT

FIELD OF THE INVENTION

The invention relates to a method of manufacturing an Al or Al alloy workpiece, such as a brazing sheet product, comprising the steps of providing an Al or Al alloy workpiece, pre-treating of the outersurface of the Al or Al alloy workpiece, and plating a metal layer comprising nickel onto the outersurface of the pre-treated workpiece. The invention also relates to a brazed assembly comprising at least one component made of the workpiece obtained by the method of this invention. The invention further relates to an aqueous plating bath for use in the method of this invention.

DESCRIPTION OF THE RELATED ART

For the purpose of this invention brazing sheet is to be understood as a core sheet, for example of aluminium or aluminium alloy, having on at least one side a brazeable aluminium alloy. Typical brazeable aluminium alloys useful as a clad layer are the Aluminium Association (AA) 4xxx-series alloys, typically having Si in the range of 2 to 18 weight %, and preferably in the range of 7 to 14%. The brazeable aluminium alloys may be coupled to the core alloy in various ways known in the art, for example by means of roll bonding, cladding, spray-forming or semi-continuous or continuous casting.

Controlled Atmosphere Brazing ("CAB") and Vacuum Brazing ("VB") are the two main processes used for industrial scale aluminium brazing. Industrial vacuum brazing has been used since the 1950's, while CAB became popular in the early 1980's after the introduction of the Nocolok (trade mark) brazing flux. Vacuum brazing is an essentially discontinuous process and puts high demands on material cleanliness. The disruption of the oxide layer present is mainly caused by the evaporation of magnesium from the clad alloy. There is always more magnesium present in the furnace then necessary. The excess magnesium condenses on the cold spots in the furnace and has to be removed frequently. The capital investment for suitable equipment is relatively high.

CAB requires an additional process step prior to brazing as compared to VB,

10

15

20

25

30

since a brazing flux has to be applied prior to brazing. CAB is essentially a continuous process in which, if the proper brazing flux is being used, high volumes of brazed assemblies can be manufactured. The brazing flux dissolves the oxide layer at brazing temperature allowing the clad alloy to flow properly. When the Nocolok flux is used the surface needs to be cleaned thoroughly prior to flux application. To obtain good brazing results the brazing flux has to be applied on the total surface of the brazed assembly. This can cause difficulties with certain types of assemblies because of their design. For example, because evaporator type heat exchangers have a large internal surface, problems can arise because of poor access to the interior. For good brazing results the flux has to adhere to the aluminium surface before brazing. Unfortunately the brazing flux after drying can easily fall off due to small mechanical vibrations. During the brazing cycle, corrosive fumes such as HF are generated. This puts a high demand on the corrosion resistance of the materials applied for the furnace.

Ideally, a material should be available that can be used for CAB but does not have the requirements and defects of the brazing flux application. Such a material can be supplied to a manufacturer of brazed assemblies and is ready to use directly after forming of the assembly parts. No additional brazing fluxing operations have to be carried out. Presently, only one process for flux-less brazing is used on an industrial scale. The material for this process can be for example standard brazing sheet made from an AA3xxx-series core alloy clad on both sides with a cladding of an AA4xxx-series alloy. Before the brazing sheet can be used the surface has to be modified in such a way that the naturally occurring oxide layer does not interfere during the brazing cycle. The method of achieving good brazing is to deposit a specific amount of nickel on the surface of the aluminium clad alloy. If properly applied, the nickel reacts with the underlying aluminium. The nickel can be applied by using a shim of nickel between the two parts to be joined or can be deposited by electroplating. When electroplating is used the adherence of the nickel should be sufficient to withstand typical shaping operations being used in for example heat exchanger manufacture.

The processes for nickel-plating in an alkaline solution of aluminium brazing sheet are known from each of US-A-3,970,237, US-A-4,028,200, US-A-4,164,454, and SAE-paper no. 880446 by B.E. Cheadle and K.F. Dockus. According to these

documents, nickel or cobalt, or combinations thereof, are most preferably deposited in combination with lead. The lead addition is used to improve the wettability of the clad alloy during the brazing cycle. An important characteristic of these plating processes is that the nickel is preferentially deposited on the silicon particles of the clad alloy. To obtain sufficient nickel for brazing on the surface, the clad alloy should contain a relatively large number of silicon particles to act as nuclei for the nickel deposition. It is believed that to obtain sufficient nucleation sites before pickling a part of the aluminium in which the silicon particles are embedded should be removed by chemical and/or mechanical pre-treatment. This is believed a necessary condition to obtain a sufficient nickel coverage to serve as nuclei for the plating action of the brazing or clad alloy. On a microscopic scale the surface of the Si-containing cladding of the brazing sheet is covered with nickel globules.

However, the use of lead for the production of a suitable nickel and/or cobalt layer on brazing sheet has several disadvantages. The plating baths for electroplating are rather complex and due to the presence of lead comprising components such as salts thereof, these baths are much more environmentally unfriendly than plating baths comprising nickel- or cobalt-components alone. The use of lead for manufacturing products, such as automotive products, is undesirable and it is envisaged that in the very near future there might possibly even be a ban on lead comprising products or products manufactured via one or more intermediate processing steps comprising lead or lead-based components. A further disadvantage of the plating bath described in US-A-4,028,200 is the considerable generation of ammonia fumes on the workshop floor due to the use of significant amounts of ammonia in the plating bath used.

SUMMARY OF THE INVENTION

5

20

25

It is an object of the present invention to provide a method of manufacturing an Al or Al alloy workpiece, comprising the steps of providing an Al or Al alloy workpiece, pretreating of the outersurface of the Al or Al alloy workpiece, and plating a metal layer comprising nickel onto the outersurface of the pre-treated Al or Al alloy

25

30

workpiece, by which method a product is obtained having good brazeability and/or weldability without the mandatory addition of lead to the nickel-layer.

It is another object of the present invention to provide a method of manufacturing a brazing sheet product comprising the steps of (a) providing a sheet comprising a core sheet coupled on at least one surface of said core sheet to an aluminium clad layer, the aluminium clad layer being made of an aluminium alloy comprising silicon in an amount in the range of 2 to 18% by weight, (b) pre-treating of the outersurface of the aluminium clad layer, and (c) plating a metal layer comprising nickel onto the outersurface of the pre-treated aluminium clad layer, by which method a brazing sheet product is obtained having good brazeability without the mandatory addition of lead to the nickel-layer.

It is a further object of the present invention to provide a method of manufacturing an Al or Al alloy workpiece, ideally a brazing sheet product, the method comprising the steps of (a) providing an Al or Al alloy workpiece, in particular a brazing sheet comprising a core sheet coupled on at least one surface of said core sheet to an aluminium clad layer, the aluminium clad layer being made of an aluminium alloy containing silicon in an amount in the range of 2 to 18% by weight, (b) pre-treating of the outersurface of the Al workpiece, in particular the aluminium clad layer, and (c) plating a metal layer comprising nickel onto the outersurface of the pre-treated workpiece, in particular of the pre-treated aluminium clad layer, by which method there is no generation of ammonia fumes originating from the plating bath when operated on an industrial scale.

It is also an object of the invention to provide an aqueous plating bath for use in the method according to the invention and which can be employed in an industrial environment.

In accordance with the invention in one aspect there is provided a method of manufacturing an aluminium or aluminium alloy workpiece, the method comprising the sequential steps of (a) providing an Al or Al alloy workpiece, (b) pre-treating of the outersurface of the aluminium workpiece, and (c) plating a metal layer comprising nickel onto said outersurface of the pre-treated aluminium or aluminium alloy workpiece. This method is characterised in that during step (c) said metal layer

WO 01/88226 PCT/EP01/04923

5

comprising nickel is deposited by electroplating both nickel and bismuth using an aqueous bath comprising a nickel-ion concentration in a range of 10 to 100 g/l and a bismuth-ion concentration in the range of 0.01 to 10 g/l.

In accordance with the invention it has been found surprisingly that the nickel layer does not need to comprise any lead as a mandatory alloying addition in order to achieve good brazeability when using the aluminium workpiece in a brazing operation. Surprisingly it has been found that equal or even better results can be obtained if bismuth is added to the nickel layer, such that said nickel layer can be kept essentially lead-free and simultaneously also in the plating bath used for the deposition of this Ni-Bi layer. By using this aqueous plating bath the need for the addition of lead has been overcome, which is a significant achievement from an environmental point of view.

10

15

20

25

30

In an embodiment of the method according to the invention this method is characterised in that during step (c) said layer comprising nickel being deposited by plating both nickel and bismuth using an aqueous bath comprising a nickel-ion concentration in a range of 20 to 70 g/l and a bismuth-ion concentration in the range of 0.02 to 5 g/l.

The nickel-ion concentration to the aqueous bath can be added via the addition of nickel chloride, nickel fluoborate, nickel sulfamate, nickel acetate or nickel sulphate. However, there is a preference to use the addition of nickel sulfate (NiSO₄.6H₂O). At a too high level of nickel salt in the aqueous bath there is the risk of the crystallisation of the salt in the solution, which might damage a continuous process. At too low levels the resultant bath becomes uneconomical due to too long plating times and low current density.

Bi-ion in the concentration set out above can be added in various ways to the aqueous bath. In theory many bismuth compounds could be used for this purpose. However, many bismuth compounds have been tried out but only a very few appear to provide reliable and reproducible results. For example the addition of bismuth acetate has been tried, but it has been found that this compound did not dissolve in the plating bath used, whereas the addition of lead acetate did not result in any problems with respect to having this compound dissolved. For example also the combination of a

bath of nickel-ions and bismuth-ions and a tartrate at a pH in the range of more than 8 resulted in the formation of an undesirable Ni containing sludge. This Ni containing sludge did not re-dissolve upon heating, indicating amongst others that Ni is unstable in the presence of a tartrate in the mentioned pH range. In accordance with the invention very good results have been obtained when Bi-ions are being added via the addition of one or more of the group consisting of bismuth carbonate (Bi₂(CO₃)₃), bismuth oxide (Bi₂O₃), bismuth citrate (BiC₆H₅O₇) and bismuth chloride (BiCl₃). Optionally some sodium hydroxide may be added also to regulate the pH of the aqueous bath. By using bismuth carbonate or bismuth oxide in the presence of nickel a suitable plating bath has been obtained which is stable at a very wide pH range. At too high levels of Bi-ion concentration in the aqueous bath the resultant deposit has a undesired high Bi-concentration. Preferably the Bi-concentration in the resultant Ni-Bi layer on the brazing sheet product is not more than 5 percent by weight, and preferably not more than 3 percent by weight. At too low levels the resultant bath becomes uneconomical due to too long plating times and low current density.

Preferably in the workpiece, in particular a brazing sheet product, the layer comprising nickel and bismuth has a thickness up to 2.0μm, preferably in the range of 0.03 to 1.0μm, and more preferably in the range of 0.05 to 0.5μm. A coating thickness of greater than 2.0μm requires a prolonged treatment time for plating, and may result in wrinkling of the molten filler material during subsequent brazing operations. A preferred thickness for this Ni-Bi containing layer is 0.3μm. Also other techniques such as roll bonding, thermal spraying, Chemical Vapour Deposition and Physical Vapour Deposition or other techniques for depositing of metal or metal alloys from a gas or vapour phase may be used.

In an embodiment of the method of the invention during step (c) said layer comprising nickel being deposited by plating both nickel and bismuth using an aqueous bath having:-

a pH in the range of 2.5 to 10, and

15

20

- comprising a nickel-ion concentration in a range of 10 to 100 g/l, and preferably in a range of 20 to 70 g/l,
- a bismuth-ion concentration in the range of 0.01 to 10 g/l, and preferably in the range of 0.02 to 5 g/l,
- 5 a citrate-ion concentration in the range of 40 to 150 g/l, and preferably in the range of 80 to 110 g/l,
 - a gluconate-ion concentration in the range of 2 to 80 g/l, and preferably in the range of 4 to 50 g/l,
- a chloride- or fluoride-ion concentration in the range of 1 to 50 g/l, and preferably in the range of 1 to 30 g/l.

This aqueous plating bath demonstrated to be operational in a very wide pH range, and can be used on industrial scale coil plating lines using a high current density, which in turn allows for fairly high line speeds. Further advantages of this plating bath are that it does not generate any ammonia fumes, it can be composed using standard and readily available chemicals, and bismuth can easily be replenished to the plating bath from a bismuth concentrate or otherwise.

Baths using the following salts have proved particularly effective, in grams per litre:-

- Nickel sulphate in a range of 45 to 450 g/l, and preferably 90 to 315 g/l,
- 20 Chloride-ion concentration in a range of 1 to 50 g/l, and preferably 1 to 30 g/l,
 - Sodium citrate in a range of 55 to 180 g/l, and preferably 110 to 150 g/l,
 - Sodium gluconate in range of 2 to 90 g/l, and preferably 5 to 55 g/l,
 - Ammonium sulphate in a range up to 270 g/l,
- Bismuth oxide in a range of 0.02 to 22 g/l, and preferably 0.05 to 11 g/l, or Bismuth carbonate in a range of 0.03 to 29 g/l, and preferably 0.06 to 14 g/l.

The addition of an ion from the group consisting of chloride and fluoride is required for inducing anode corrosion. A suitable source of chloride-ion concentration can be done by the addition of nickel chloride ($NiCl_2.6H_2O$) in a range of up to 415 g/l, and preferably in a range up to 250 g/l.

10

20

(H⁺) or (OH) can be added to regulate the pH in a range of 2.5 to 10. The use of ammonium hydroxide should preferably be avoided in view of the generation of ammonia fumes.

Optionally for reducing stress in the deposit layer comprising the Ni and Bi an ammonium-ion concentration in a range up to 40 g/l, and preferably in range of 1 to 25 g/l, or a triethenalamine-ion concentration in a range of up to 40 g/l, or combinations thereof, or other equivalent components may be added to the aqueous bath. Any soluble ammonium salt can be used as a source of NH₄⁺.

The plating bath used in the method according to the invention can operate in a wide pH range of 2.5 to 10, and preferably in the range of 4 to 8, without affecting the properties of the bath and without dissolving the aluminium clad layer or any other metal layer thereon. If the aluminium workpiece, such as a brazing sheet product having an aluminium clad layer, is provided with a thin intermediate zinc layer, e.g. by means of a zincate treatment via direct or immersion plating, prior to the plating of the Ni-Bi layer, the pH is preferably in the range of 5 to 8, and more preferably in the range of 5.4 to 7.5. In the embodiment of an applied intermediate layer comprising zinc, the layer has a thickness up to 0.5µm, more preferably up to 0.3µm (300nm), and most preferably in the range of 0.01 to 0.15µm (10-150nm). In the best results obtained a thickness of about 30nm has been used. A coating thickness of greater than 0.5µm requires a prolonged treatment time, e.g. for displacement plating, and is thought to have no further advantages for improving the adhesion. Instead of zinc also tin may be used.

The method according to the invention is preferably employed using a plating bath having a temperature in the rang of 30 to 70°C, and more preferably in the range of 40 to 65°C. In this temperature range the ion-mobility increases and there is no need to cool the plating bath to compensate for the heat generation during plating.

The invention further relates to an aqueous bath for the electro-deposition of a layer comprising both nickel and bismuth on a brazing sheet product having:-

a pH in the range of 2.5 to 10, and

20

25

30

- comprising a nickel-ion concentration in the range of 10 to 100 g/l, and preferably in a range of 20 to 70g/l,
- a bismuth-ion concentration in the range of 0.01 to 10 g/l, and preferably in a range of 0.02 to 5 g/l,
- 5 a citrate-ion concentration in the range of 50 to 150 g/l, and preferably in a range of 80 to 110 g/l,
 - a gluconate-ion concentration in the range of 2 to 80 g/l, and preferably in a range of 4 to 50 g/l,
 - a chloride- or fluoride-ion concentration in the range of 1 to 50 g/l, and preferably in a range of 1 to 30 g/l.

This aqueous plating bath demonstrated to be operational in a very wide pH range, and at a wide temperature range, and further can be used on industrial scale coil plating lines using a high current density, which in turn allows for fairly high line speeds. Further advantages of this plating bath are that it does not generate any ammonia fumes, it can be composed used standard and readily available chemicals, and the bismuth concentration can easily be replenished to the plating bath from a bismuth concentrate or otherwise.

The adhesion of the layer comprising both nickel and bismuth to the aluminium workpiece, such as the cladding of a brazing sheet product, is fairly good, but may be further improved by a proper pre-treatment of the outersurface of the aluminium workpiece on which the Ni-Bi layer is being deposited, such as the clad layer of a brazing sheet product. The pre-treatment comprises a preliminary cleaning step during which the surface is made free from grease, oil, or buffing compounds. This can be accomplished in many ways, and can be accomplished amongst others by vapour degreasing, solvent washing, or solvent emulsion cleaning. Also a mild etching may be employed. Following the preliminary cleaning, the surface should preferably be conditioned. Several methods can be applied successfully, such as, but not limited thereto:-

(a) acid desmutting in an solution comprising nitric acid (typically 25-50%), optionally in combination with a fluoride and/or chromic acid and/or sulphuric acid. Suitable sources for the fluoride can be, for example, hydrofluoric acid or ammonium

bifluoride, see also e.g. "The surface treatment and finishing of aluminium and its alloys", by S. Wernick *et al.*, ASM international, 5th edition, 1987, Vol.1, pp.181-182.

- (b) mechanical preparation such as polishing, abrasion or brushing. These may also be applied while the surface is in contact with a lower alcohol, such as for example isopropanol, see e.g. also US-A-4,388,159.
 - (c) alkaline etching, see e.g. "The surface treatment and finishing of aluminium and its alloys", by S. Wernick *et al.*, ASM international, 5th edition, 1987, Vol.1, pp.191-203.
- 10 (d) aqueous detergent cleaning;
 - (e) anodic oxidation, see e.g. "The surface treatment and finishing of aluminium and its alloys", by S. Wernick et al., ASM International, 5th edition, 1987, Vol.2, pp.1006 ff.
 - (f) electrograining.

30

- 15 (g) pre-treatments described in for example US-A-4,741,811, US-A-5,245,847, US-A-5,643,434.
 - (h) immersion processes such as zincate and stannate, see "The surface treatment and finishing of aluminium and its alloys", by S. Wernick *et al.*, ASM international, 5th edition, 1987, Vol.2, Chapter 14 and 15.
- Also combinations of one or more conditioning steps set out above can be applied successfully.

In an embodiment the Al or Al alloy workpiece is an aluminium alloy sheet or aluminium alloy wire or aluminium alloy rod. Although various aluminium alloys may be applied, e.g. those within the Aluminium Association (AA)3xxx- and AA6xxx-series aluminium alloys. Particular suitable aluminium alloys are those within the AA4xxx-series alloys, typically having Si as the most important alloying element in the range of 2 to 18% by weight, more preferably 7 to 14% by weight. Other alloying elements may be present to improve specific properties, the balance is made by iron up to 0.8%, and impurities each up to 0.05 wt.%, total up to 0.20 wt.%, and aluminium. An AA4xxx-series aluminium alloy sheet can be plated with Ni-Bi alloy in accordance with the method of the invention, and may be employed in

WO 01/88226 PCT/EP01/04923

11

subsequent brazing operations, in particular in an inert atmosphere brazing (CAB) operation in the absence of a brazing-flux material. Also aluminium alloy wire or rods being made of an AA4xxx-series alloy may be plated with a Ni-Bi layer, and subsequently employed in brazing operations in particular in an inert atmosphere brazing (CAB) operations in the absence of a brazing-flux material, and may also be employed as weld filler wire or weld filler rod in a welding operation.

In a preferred embodiment the aluminium alloy workpiece is a brazing sheet product comprising a core sheet coupled at least one surface of said core sheet to n aluminium clad layer, the aluminium clad layer being made of an aluminium AA4xxx-series alloy comprising silicon in the range of 2 to 18% by weight, preferably in the range of 7 to 14%. In an embodiment of the aluminium brazing sheet product, the core sheet is made of an aluminium alloy, in particular those within the AA3xxx, AA5xxx, or AA6xxx-series alloys.

10

15

20

25

The invention further provides a brazed assembly comprising at least one component made of the aluminium alloy workpiece, in particular a brazing sheet product, obtained by the method in accordance with the invention described above. Preferably the resultant brazing sheet product has a Ni-Bi layer comprising Bi, by weight percent, in a range of up to 5%, preferably not more than 3%, and more preferably up to 1.0%, and most preferably in a range of 0.01 to 0.05%.

In an embodiment there is provided a brazed assembly wherein at least one of the parts to be joined by brazing is made of the aluminium alloy workpiece, in particular a brazing sheet product, produced by the method in accordance with the invention described above, and at least one other part is made of steel, aluminized steel, stainless steel, plated or coated stainless steel, bronze, brass, nickel, nickel alloy, titanium, or plated or coated titanium.

In a further aspect of the invention there is provided in a method of manufacturing a brazed assembly using the aluminium alloy workpiece, in particular a brazing sheet product, produced by the method in accordance with this invention, comprising the steps of:-

- (a) shaping or forming parts of which at least one is made from the aluminium alloy workpiece, ideally a brazing sheet product, obtained by the method in accordance with this invention as set out above:
- (b) assembling the parts into the assembly;
- 5 (c) brazing the assembly under a vacuum or in an inert atmosphere (CAB) in the absence of a brazing-flux at elevated temperature for a period long enough for melting and spreading of the molten filler alloy;
 - (d) cooling the brazed assembly to below 100°C. The cooling rate may be in the range of typical brazing furnace cooling rates. Typical cooling rates are cooling rates of at least 10 °C/min or more.

In dependence on the material, particularly aluminium alloy, of the cores sheet the process may include the further processing step (e) of ageing of the brazed and cooled assembly in order to optimise the mechanical and/or corrosion properties of the resultant assembly. The use of the brazing sheet product obtained from the method according to the invention set out above has been found to result in a lower brazing temperature by at least 10°C. This reduced brazing temperature allows a significant reduction of the industrial scale processing time of a complete brazing cycle, typically a time reduction of 15% or more has been found.

In an embodiment of the method of manufacturing a brazed assembly in step (a) at least one of the parts to be joined by brazing is made of the brazing sheet product produced by the method in accordance with the invention described above, and at least one other part is made of steel, aluminized steel, stainless steel, plated or coated stainless steel, bronze, brass, nickel, nickel alloy, titanium, or plated or coated titanium.

25

30

10

15

20

EXAMPLE.

On a laboratory scale of testing aluminium brazing sheets manufactured from an AA3003 core alloy clad on both sides with an AA4045 clad alloy, and having a total thickness of 0.5 mm and a clad layer thickness of 50 microns each, was treated using the following sequential process steps:

cleaning by immersion for 180 sec. in ChemTec 30014 (a commercial available

20

25

30

alkaline (etch) degreaser), followed by rinsing;

- alkaline etching for 20 sec. in ChemTec 30203 (a commercial available alkaline etch cleaner) at ambient temperature, followed by rinsing;
- desmutting for 4 sec. in an acidic oxidizing bath, typically 25-50 vol.% nitric acid, comprising ChemTec 11093 (a commercial available pickle activator) at ambient temperature, followed by rinsing;
 - nickel electroplating, and rinsing.

The nickel plating bath used has the composition of Table 1 and having a pH of 5.5.

The Bi-ion concentration has been added to the plating bath using a Bi-ion concentrate of 160 g/l sodium hydroxide, 300 g/l sodium gluconate and 110 g/l bismuth oxide. The bismuth oxide could have been replaced also by bismuth carbonate. The electroplating of a Ni-Bi layer was performed at 57°C using three different current densities and plating times. The composition of the resultant layer was measured using ICP and is given in Table 2. ICP stands for Induced Coupled Plasma. The results given are for the sum of both plated sides.

For comparison similar brazing sheet material has been Ni-plated with Ni-Pb. The plating bath comprised 50 g/l nickel sulphate, 50 g/l nickel chloride, 100 g/l/ sodium citrate, 1 g/l lead acetate, and 75 ml/l ammonium hydroxide (30%). The plating conditions at 26°C were such that a plating time of 200s resulted in a nickel-lead plated layer of 2.0 microns using a current density of 3 A/dm². Due to the presence of ammonium-hydroxide ammonia fumes are being generated.

The nickel plated specimens have been tested for adhesion using the Erichsen dome test (5 mm), and the T-bend test. A value assessment is then given to the adhesion where: (-) = poor, (\pm) = fair, and (+) = good. The results are given in Table 2. Further the brazeability has been assessed. On a laboratory scale of testing the brazing tests were carried out in a small quartz furnace. Small coupons of 25 mm x 25 mm were cut from the nickel-plated sheets. A small strip of an AA3003 alloy measuring 30 mm x 7 mm x 1 mm was bent in the centre to an angle of 45° and laid on the coupons. The strip-on-the-coupon samples were heated under flowing nitrogen, with heating from room temperature to 580°C, dwell time at 580°C for 1 minute, cooling from 580°C to room temperature. The brazing process was judged on possible

10

15

20

25

formation of wrinkles, capillary depression and fillet formation. An overall assessment was given where: (-) = poor brazeability, $(-/\pm)$ = fair brazeability, (\pm) = good brazeability, and (+) = excellent brazeability. The results obtained are summarised in Table 2.

This example shows how an electroplated nickel layer containing Bi, but comprising no Pb, may be applied and resulting in a product having at least a fair adhesion of the nickel layer and excellent brazeability. Normally bismuth is not easily maintained in a stable nickel-plating solution without sludge formation.

It will be apparent to the skilled person that the Ni-Bi plating operations according to the invention may be applied also on one or both sides of an aluminium alloy sheet or strip made of an AA4xxx-series aluminium alloys, which aluminium alloy sheet is not being provided with a core sheet to form a brazing sheet product. Such a Ni-Bi plated sheet or strip, typically having a gauge in the range of up to 3mm, preferably in the range of 0.05 to 2mm, may be employed also in a brazing operation as set out in this example. A similar approach can be used for plating aluminium alloy wires or rods. Such Ni-Bi plated wires or rods may be employed in a brazing operation as set out in this example or used as filler material in a welding operation, such as for laser welding operations.

The current efficiency of Ni deposition appears to be 100%.

The samples plated with Ni-Bi at 3 A/dm⁻² showed some black spots, but the samples plated at the higher current densities have excellent appearance. About 0.5 g.m⁻² bismuth was deposited. The bismuth content of the deposited alloy layer can easily be varied, e.g. by lowering the bismuth concentration in the plating bath, to give a lower Bi content.

This plating bath has many advantages compared to the standard known Pb-containing baths:-

no ammonia fumes

more practical operating temperatures, typically 40 to 70°C

high current density

bismuth can easily be replenished to the plating bath.
Further, standard chemicals were employed.

Table 1

Compound	Concentration [g/l]		
Nickel sulphate	142		
Ammonium sulphate	34		
Nickel chloride	30		
Sodium citrate	140		
Sodium gluconate	30		
Bismuth ions	1		

Table 2

Invention	Current	Time	Ni	Bi	Adhesion	Brazeability
	Density [A/dm ²]	[sec.]	[g/m ²]	[g/m²]		
yes	3	50	9.1	0.66	±	+
yes	6	25	10.4	0.50	±	+
yes	10	15	9.5	0.46	±	+
no	3	50	9.4	-	±	+

Having now fully described the invention, it will be apparent to one of ordinary skill in the art that many changes and modifications can be made without departing from the spirit or scope of the invention as herein described.

CLAIMS

- A method of manufacturing an Al or Al alloy workpiece, which method comprises the steps of (a) providing an Al or Al alloy workpiece, (b) pretreating of the outersurface of the Al or Al alloy workpiece, and (c) plating a metal layer comprising nickel onto said outersurface of the Al or Al alloy workpiece, characterised in that during step (c) said metal layer comprising nickel is deposited by electroplating both nickel and bismuth using an aqueous bath comprising a nickel-ion concentration in a range of 10 to 100 g/l and a bismuth-ion concentration in the range of 0.01 to 10 g/l.
 - 2. A method according to claim 1, wherein during step (c) said metal layer comprising nickel being deposited by plating both nickel and bismuth using an aqueous bath having a pH in the range of 2.5 to 10, and comprising

a nickel-ion concentration in a range of 10 to 100 g/l,

- a bismuth-ion concentration in the range of 0.01 to 10 g/l,
- a citrate-ion concentration in the range of 40 to 150 g/l,
- a gluconate-ion concentration in the range of 2 to 80 g/l,
- a chloride- or fluoride-ion concentration in the range of 1 to 50 g/l.

20

- 3. A method according to claim 1 or 2, wherein the aqueous bath and the resultant metal layer comprising nickel are essentially lead-free.
- 4. A method according to any of claims 1 to 3, wherein said layer comprising nickel has a thickness of not more than 2.0 microns, and preferably in the range of 0.03 to 1.0 micron.
 - 5. A method according to any one of claims 1 to 4, wherein the Bi-ion concentration results from the dissolution of one or more from the group consisting of bismuth carbonate, bismuth oxide, bismuth citrate and bismuth chloride.

- 6. A method according to any one of claims 1 to 5, wherein the temperature of the aqueous bath during plating is in the range of 30 to 70 °C.
- 5 7. A method according to any one of claims 1 to 6, wherein the workpiece is aluminium alloy sheet or aluminium alloy wire or rod.
 - 8. A method according to claim 7, wherein the aluminium alloy comprises silicon as an alloying element in the range of 2 to 18% by weight.
- 9. A method according to claim 7, wherein the workpiece is a brazing sheet product comprising a core sheet coupled on at least one surface of said core sheet to an aluminium clad layer, the aluminium clad layer being made of an aluminium alloy comprising silicon in an amount in the range of 2 to 18% by weight, and wherein during step (b) at least the outersurface of the aluminium clad alloy is being pre-treated.
- 10. An aqueous bath for the electrodeposition of a layer of nickel and bismuth on an Al or Al alloy workpiece, having a pH in the range of 2.5 to 10, and comprising
 20 a nickel-ion concentration in a range of 10 to 100 g/l,
 a bismuth-ion concentration in the range of 0.01 to 10 g/l,
 a citrate-ion concentration in the range of 50 to 150 g/l,
 a gluconate-ion concentration in the range of 2 to 80 g/l,
 a chloride- or fluoride-ion concentration in the range of 1 to 50 g/l.
 - 11. Use of the aqueous bath of claim 10 for the manufacturing of Ni-plated products, preferably a brazing sheet product, for use in a brazing operation.
- 12. An assembly of components joined by brazing, at least one said components being an Al or Al alloy workpiece produced by the method in accordance with any one of claims 1 to 9.

- 13. Method of manufacturing an assembly of brazed components, comprising the steps of:-
 - (a) shaping parts of which at least one is made from an Al or Al alloy workpiece obtained by the method according to any one of claims 1 to 9;
 - (b) assembling the parts into the assembly;
 - (c) brazing the assembly in an inert atmosphere in the absence of a brazingflux at elevated temperature for a period long enough for melting and spreading of the molten filler;
- 10 (d) cooling the brazed assembly to below 100°C.

(19) World Intellectual Property Organization International Bureau





(43) International Publication Date 22 November 2001 (22.11.2001)

PCT

(10) International Publication Number WO 01/088226 A3

(51) International Patent Classification⁷: B23K 35/28, C25D 5/44

C25D₃/56,

(21) International Application Number: PCT/EP01/04923

(22) International Filing Date: 27 April 2001 (27.04.2001)

(25) Filing Language:

English

(26) Publication Language:

English

(30) Priority Data: PCT/EP00/04608 EP 00203304.1

EP 01200628.4

18 May 2000 (18.05.2000) EP 25 September 2000 (25.09.2000) EP 21 February 2001 (21.02.2001) EP

(71) Applicant (for all designated States except US): CO

(71) Applicant (for all designated States except US): CORUS ALUMINIUM WALZPRODUKTE GMBH [DE/DE]; Carl-Spaeter-Strasse 10, 56070 Koblenz (DE).

(72) Inventors; and

(75) Inventors/Applicants (for US only): MOOIJ, Joop, Nicolaas [NL/NL]; Walstro 56, NL-1902 JP Castricum (NL). WITTEBROOD, Adrianus, Jacobus [NL/NL]; Vederkruid 8, NL-1991 HB Velserbroek (NL). WI-JENBERG, Jacques, Hubert, Olga, Joseph [NL/NL]; Bogortuin 135, NL-1019 PE Amsterdam (NL). (74) Agent: HERMAN DE GROOT, Johan, Willem; Corus Technology BV, P.O. Box 10000, NL-1970 CA IJmuiden (NL).

(81) Designated States (national): AE, AG, AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, BZ, CH, CN, CR, CU, CZ, DE, DK, DM, DZ, EE, ES, FI, GB, GD, GE, GH, GM, HR, HU, ID, IL, IN, IS, JP, KE, KG, KP, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MA, MD, MG, MK, MN, MW, MX, MZ, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, TZ, UA, UG, US, UZ, VN, YU, ZA, ZW.

(84) Designated States (regional): ARIPO patent (GH, GM, KE, LS, MW, MZ, SD, SL, SZ, TZ, UG, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), European patent (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE, TR), OAPI patent (BF, BJ, CF, CG, CI, CM, GA, GN, GW, ML, MR, NE, SN, TD, TG).

Published:

with international search report

(88) Date of publication of the international search report: 18 July 2002

For two-letter codes and other abbreviations, refer to the "Guidance Notes on Codes and Abbreviations" appearing at the beginning of each regular issue of the PCT Gazette.

(54) Title: METHOD OF MANUFACTURING AN ALUMINIUM PRODUCT

(57) Abstract: The invention relates to a method of manufacturing an Al or alloy workpiece, comprising the steps of (a) providing an AlAl alloy workpiece, (b) pre-treating of the outersurface of the Al or Al alloy workpiece, and (c) plating a metal layer comprising nickel onto said outersurface of the pre-treated Al or Al alloy workpiece, and wherein during step (c) said metal layer comprising nickel is deposited by electroplating both nickel and bismuth using an aqueous bath comprising a nickel-ion concentration in a range of 10 to 100 g/l and a bismuthion concentration in the range of 0.01 to 10 g/l. The invention further relates to an aqueous plating bath for use in the method of this invention.

INT RNATIONAL SEARCH REPORT

Inter anal Application No PCT/EP 01/04923

A. CLASSIFICATION OF SUBJECT MATTER IPC 7 C25D3/56 B23K B23K35/28 C25D5/44 According to International Patent Classification (IPC) or to both national classification and IPC Minimum documentation searched (classification system followed by classification symbols) C25D B23K IPC 7 Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched Electronic data base consulted during the international search (name of data base and, where practical, search terms used) EPO-Internal, WPI Data, PAJ, CHEM ABS Data C. DOCUMENTS CONSIDERED TO BE RELEVANT Citation of document, with indication, where appropriate, of the relevant passages Relevant to daim No. X US 3 970 237 A (DOCKUS KOSTAS F) 1,3,4, 20 July 1976 (1976-07-20) 6-9,12,cited in the application 13 the whole document Α US 4 785 137 A (SAMUELS GEORGE J) 2-6,1015 November 1988 (1988-11-15) column 6, line 51 -column 7, line 27; example 2; tables 1,2 Further documents are listed in the continuation of box C. Patent family members are listed in annex. Special categories of cited documents: T later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the investment. *A* document defining the general state of the art which is not considered to be of particular relevance invention "E" earlier document but published on or after the international "X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to filing date document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified) involve an inventive step when the document is taken alone "Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such docu-ments, such combination being obvious to a person skilled document referring to an oral disclosure, use, exhibition or other means document published prior to the international filing date but later than the priority date claimed in the art. "&" document member of the same patent family Date of the actual completion of the international search Date of mailing of the international search report 13 February 2002 01/03/2002 Name and mailing address of the ISA Authorized officer European Patent Office, P.B. 5818 Patentiaan 2 NL - 2280 HV Rijswijk Tel. (+31-70) 340-2040, Tx. 31 651 epo ni, Fax: (+31-70) 340-3016 Van Leeuwen, R

INT'RNATIONAL SEARCH REPORT

Inter anal Application No
PCT/EP 01/04923

C (Ca=N=-	NOO! IMENTS CONSIDERED TO SEE THE	PCT/EP 01	/ U4923
Category °	ation) DOCUMENTS CONSIDERED TO BE RELEVANT Citation of document, with indication, where appropriate, of the relevant passages	·	Delever to about
oulcgory	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.	
A	DATABASE CA 'Online! CHEMICAL ABSTRACTS SERVICE, COLUMBUS, OHIO, US; POVETKIN, V. V. ET AL: "Electroplating and properties of nickel-bismuth alloys in a Trilon bath" retrieved from STN	2,5,10	
	Database accession no. 103:168695 CA XP002190053 abstract & ZASHCH. MET. (1985), 21(4), 643-5 , 1985,		
			•
		5	
*			

INTERNATIONAL SEARCH REPORT

information on patent family members

Inter nathmal Application No PCT/EP 01/04923

Patent document cited in search report		Publication date		Patent family member(s)	Publication date
US 3970237	Α	20-07-1976	US	4028200 A	07-06-1977
			AU	6193973 A	01-05-1975
			ΒE	806897 A1	06-05-1974
			CA	1020816 A1	15-11-1977
			CA	1034074 A2	04-07-1978
			DE	2354588 A1	16-05-1974
			DE	2365799 A1	06-05-1976
			FR	2205581 A1	31-05-1974
			GB	1444179 A	28-07-1976
		•	ΙT	999233 B	20-02-1976
			JP	1158651 C	25-07-1983
•			JP	49134550 A	25-12-1974
			JP	57050879 B	29-10-1982
			MX	143837 A	24-07-1981
			NL	7314947 A	09-05-1974
			NO	135943 B	21-03-1977
			NO	760065 A	08-05-1974
			NO	141635 B	07-01-1980
			NO	761333 A	08-05-1974
			NO	137455 B	21-11-1977
			SE	7705860 A	18-05-1977
			ZA	7308183 A	30-04-1975
US 4785137	Α	15-11-1988	ΑU	4184685 A	07-11-1985
			EP	0160236 A1	06-11-1985
			JP	60255947 A	17-12-1985
			US	4633050 A	30-12-1986